



Local atomic and electronic structure of tungsten ions in AWO_4 crystals of scheelite and wolframite types

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Abstract

X-ray absorption spectroscopy was used to study the local atomic and electronic structure of tungsten ions in polycrystalline scheelite CaWO_4 and wolframite-type ZnWO_4 and NiWO_4 . The W L_1 - and L_3 -edges X-ray absorption near edge structure (XANES) signals suggest tetrahedral coordination of tungsten ions in CaWO_4 and strongly distorted octahedral coordination in ZnWO_4 and NiWO_4 . Accurate analysis of the W L_3 -edge extended X-ray absorption fine structure (EXAFS) signals by the regularization procedure was performed to reconstruct the radial distribution functions within the first coordination shell around tungsten atoms in AWO_4 crystals and polycrystalline WO_3 , which was utilized for comparison. For the local environment around tungsten atoms in CaWO_4 , we found very good agreement with known crystallographic data, suggesting the presence of four oxygen atoms at $\sim 1.79 \text{ \AA}$. The results obtained for ZnWO_4 and NiWO_4 show similarity of their local atomic structures: the oxygen atoms forming $[\text{WO}_6]$ octahedra can be divided into two groups — four atoms at $1.84 \pm 0.01 \text{ \AA}$ and two atoms at $2.13 \pm 0.01 \text{ \AA}$. © 2001 Elsevier Science Ltd. All rights reserved.

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1. Introduction

AWO_4 compounds are important materials due to their use as scintillator detectors, photoanodes, solid-state laser hosts and in optical fibre applications (Galasso, 1970; Kaminskii, 1981; Wang et al., 1992). They crystallise in scheelite- or wolframite-type structures. The scheelite structure (Gürmen et al., 1971) may be regarded as a cubic close-packed array of A^{2+} and $[\text{WO}_4]^{2-}$ units with the coordination numbers of 8 and 4 oxygen atoms for the A and W cations, respectively, whereas the wolframite structure (Wyckoff, 1963) may be described as made up of hexagonally close-packed oxygens with certain octahedral sites filled by A and W cations in an ordered way (Sleight, 1972).

The type of the tungstate structure depends crucially on the ionic radius of the A^{2+} cation (Sleight, 1972): for small A^{2+} cations (such as Mg^{2+} and 3d ions) the wolframite

structure is preferable, but in the case of Ca^{2+} , Ba^{2+} and Sr^{2+} the scheelite structure becomes favorable. In addition, the transition from scheelite (s) to wolframite (w) structure can be stimulated by a high pressure—about 1.2 GPa for CaWO_4 and 5 GPa for BaWO_4 (Kawada et al., 1974; Nicol and Durana, 1971). From the long-range point of view, it is accompanied by an abrupt increase in a_w , c_w or a_s lattice constants and an abrupt decrease in b_w or c_s (Sleight, 1972). From the short-range point of view, a shift of the W cation from the center of the $[\text{WO}_4]$ tetrahedron towards the center of the $[\text{WO}_6]$ octahedron occurs during the transition.

The local structure of tungsten ions can be successfully studied by X-ray absorption spectroscopy. In this work, we present recent results on the local atomic and electronic structure of tungsten ions, obtained from the analysis of the W L_1 - and L_3 - edges XANES (X-ray absorption near edge structure) and EXAFS (extended X-ray absorption fine structure) spectra in CaWO_4 , NiWO_4 and ZnWO_4 . A set of structural parameters, determined from the modeling of the EXAFS signals, will be discussed in connection with

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known data. The analysis of the XANES spectra, related to the conduction band states, will be presented within an ab initio full-multiple scattering formalism.

2. Experimental details and data analysis

Polycrystalline NiWO₄ and ZnWO₄ powders were prepared using thermal treatment in air at 700°C of amorphous powders, obtained by co-precipitation at room temperature (RT) from nickel or zinc nitrate and sodium tungstate aqueous solutions. Polycrystalline CaWO₄ and WO₃ powders were commercial products. The single-phase structure of all compounds was confirmed by X-ray powder diffraction.

The RT X-ray absorption spectra of the W L₁- and L₃-edges were recorded at the LURE DCI storage ring (Orsay, France), operated at the energy 1.85 GeV and a maximum stored current 316 mA. A standard transmission scheme with a Si(331), Si(311) and Si(111) double-crystal monochromators and two ion chambers containing argon/air was used. The energy resolution was about 1–2 eV. The X-ray absorption spectra were analysed following a standard method using the EDA software package (Kuzmin, 1995, 1997). The X-ray absorption coefficient $\mu(E)$ was calculated as $\mu(E) = \ln(I_0/I)$ using the measured intensities of the incident $I_0(E)$ and transmitted through the sample $I(E)$ X-rays. Then, the EXAFS signal $\chi(k)$ was obtained as $\chi(k) = (\mu - \mu_b - \mu_0)/\mu_0$, where μ_b is the background contribution, μ_0 is the embedded-atomic contribution and $k = [(2m_e/\hbar^2)(E - E_0)]^{1/2}$ is the photoelectron wavenumber (m_e is the electron mass, \hbar is the Planck's constant and E_0 is the energy, defining the zero photoelectron wavenumber value ($k = 0$)) (Kuzmin and Purans, 2000). The Fourier transforms of the EXAFS signals were calculated as $FT(R) = (2/\pi)^{1/2} \int_{\chi} (k)k^2 W(k) \exp(-2ikR) dk$, where $W(k)$ is the “window”-function.

The model-independent regularisation method (Kuzmin, 1997; Kuzmin and Purans, 2000) was used to reconstruct the radial distribution functions (RDFs) $G_{W-O}(R)$ within the first coordination shell of tungsten. The RDF $G_{W-O}(R)$ corresponds to the number of oxygen atoms located in the spherical shell around the tungsten between R and $R + \delta R$, so that the integration of $G_{W-O}(R)$ over the first shell range gives the coordination number. The scattering amplitude and phase shift functions, required in the EXAFS calculations, were extracted from the EXAFS signal for CaWO₄, assuming the following values of structural parameters (Gürmen et al., 1971): the coordination number of tungsten $N = 4$, the average interatomic distance $R(W-O) = 1.785 \text{ \AA}$ and the Debye–Waller factor $\sigma^2 = 0.002 \text{ \AA}^2$.

XANES signals were modelled by the ab initio full-multiple-scattering (FMS) FEFF8 code (Ankudinov et al., 1998). The clusters having the appropriate crystal structure and a size up to 8 shells around the absorber (tungsten atom) were used. The self-consistent-field (SCF) cluster potential was utilized, including the complex energy

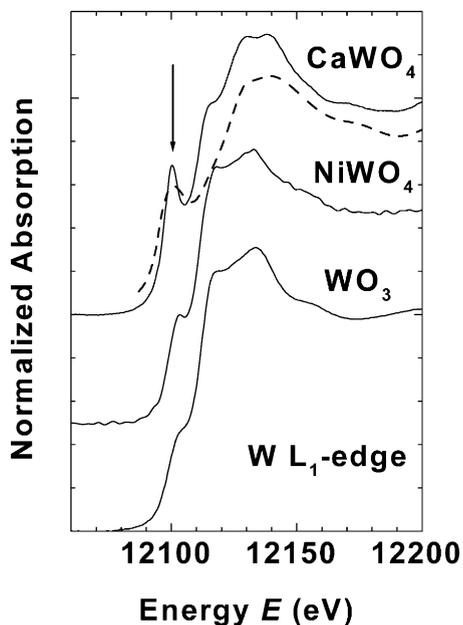


Fig. 1. Experimental W L₁-edge XANES spectra (solid lines) in CaWO₄, NiWO₄ and WO₃. The FMS calculation of the XANES signal for a cluster of 8 coordination shells around tungsten atom in CaWO₄ is shown by dashed line. The pre-edge peak, corresponding to the 2s(W) → 5d(W)+2p(O) transition, is indicated by arrow.

dependent Dirac-Hara-Hedin-Lundqvist exchange-correlation part (Ankudinov et al., 1998).

3. Results and discussion

The XANES signals at the W L₁-edge are shown in Fig. 1. They represent considerable interest, because the pre-edge peak (indicated by the arrow) is sensitive to the local symmetry of the tungsten sites (Kuzmin and Purans, 1993). The intensity of the pre-edge peak is the largest in the case of tetrahedral coordination and becomes smaller for the octahedral environment. The slightly higher intensity of the pre-edge peak in NiWO₄ compared to WO₃ suggests a stronger distortion of the [WO₆] octahedra in the former. Theoretical calculations (dashed line in Fig. 1, shown only for CaWO₄) are able to reproduce the experimental signal, however a large 8 shells cluster should be considered, and a problem to reproduce the shoulder at ~12115 eV, attributed to the 2s(W) → 6p(W) transition, exists.

The Fourier transforms (FTs) of the experimental EXAFS signals $\chi(k)k^2$ are shown in Fig. 2. The first peak at ~1.5 Å corresponds to the first coordination shell of tungsten atoms, built of oxygen atoms. The group of peaks at ~3 Å are mainly due to the second and third shell, composed of tungsten and A²⁺ atoms. High intensity of this peak for NiWO₄ and ZnWO₄ can be attributed to the presence of

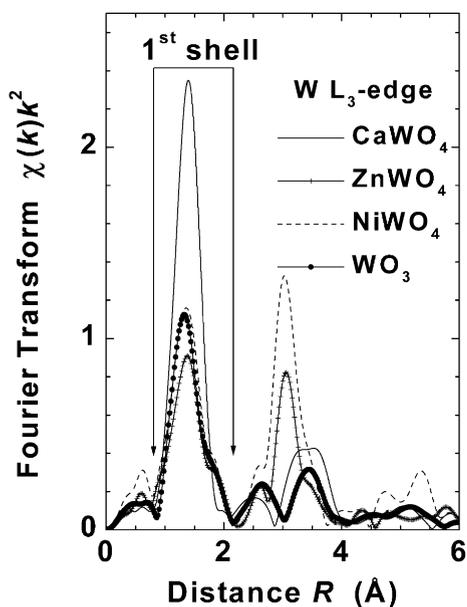


Fig. 2. Fourier transforms of the experimental W L_3 -edge EXAFS $\chi(k)k^2$ signals. The region of the first coordination shell of tungsten atoms is shown. Note that the positions of peaks do not correspond to the true crystallographic values due to the phase shift contribution in the EXAFS signal.

the so-called “multiple-scattering” contributions (Kuzmin et al., 1993). To determine structural parameters for the first shell in each compound, the peaks at ~ 1.5 Å were singled out by back-FT, and thus obtained EXAFS signals were best-fitted using the model-independent regularisation approach (Kuzmin, 1997; Kuzmin and Purans, 2000). The reconstructed radial distribution functions (RDFs) are given in Fig. 3. The shape of the RDFs suggests strong distortion of the $[WO_6]$ octahedra in tungstates and together with complementary data from Raman spectroscopy (Kuzmin et al., 2001) indicates a preferential bonding of tungsten ions to the group of four nearest oxygen atoms at 1.84 ± 0.01 Å.

4. Summary and conclusions

Local atomic and electronic structures of tungsten ions in polycrystalline scheelite $CaWO_4$ and wolframite-type $ZnWO_4$ and $NiWO_4$ were studied by X-ray absorption spectroscopy on the W L_1 - and L_3 -edges. The analysis of both XANES and EXAFS signals allows to conclude that the tungsten atoms have regular tetrahedral coordination with the W–O distances about 1.79 Å in $CaWO_4$, but they have very distorted octahedral coordination with four short (at 1.84 Å) and two long (2.13 Å) W–O distances in $ZnWO_4$ and $NiWO_4$. Such difference in the local environment of the tungsten atoms agrees well with the one, expected upon the transition from scheelite to wolframite structure.

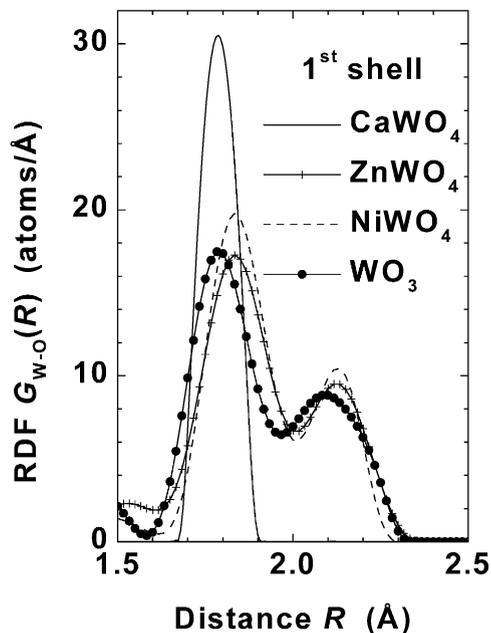


Fig. 3. Radial distribution function (RDF) $G_{W-O}(R)$ for the first coordination shell of tungsten atoms.

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